

**A composition comprising a conductive polymer in colloidal form and carbon**

**[Technical Field to which the Invention Pertains]**

The present invention relates to conductive materials comprising (intrinsically) conductive polymers and carbon materials, methods for their manufacture and use for high-capacity electrical double layer capacitors to be utilized in various electronic apparatuses, power supplies and the like.

**[Prior Art]**

Conductive Materials are known and used in many different forms and applications. Conductive materials based on carbon are available in several different physical and chemical morphologies, forms and compositions. Pure or mainly pure carbon is available in the form of carbon black (which contains also mainly oxygen based impurities, graphite (pure C), carbon nanotubes and fullerenes and the like. Another

group of conductive materials are (intrinsically) conductive polymers which have found some first applications.

These two groups of conductive materials have at least one property in common, the conductivity. Other properties may be exclusive to the one or the other group of materials, or may vary widely within the respective group. For example, the particle size of graphite is in the range of several to several tens of microns, while that of fullerenes is in the range of angstroms. The specific surfaces of carbon black and carbon nanotubes exhibit high values up to about 1000 m<sup>2</sup>/g while that of graphite is in the range of a few m<sup>2</sup>/g. Polyaniline, one of the representatives of conductive polymers, is characterized by a rich redox chemistry, PEDOT (polyethylenedioxythiophene), also termed PEDT, has a moderately or poorly reproducible redox chemistry, while graphite or carbon black show no reversible redox chemistry.

The combination of conductive polymers like polyaniline, polyethylenedioxythiophene, polypyrroles or their derivatives with carbonaceous materials (carbon black, graphite, carbon nanotubes and fullerenes) has been tried in the past. While simple mixtures of said materials do not offer any significant or reproducible advantage, and hence did not find any commercial or technological applications, chemical processes for the combination of conductive polymers and carbon black or conductive polymers and carbon nanotubes have been widely studied. For instance, the company Eeonyx has offered carbon black on which surface polyaniline was polymerised as a developmental product in the market ("Eeonomer"), cf. G. Du, A. Epstein, K. Reimer, presentation on the March 1996 Meeting of the American Physical Society, Session M23, presentation M 23.09. The laboratory chemicals supplier Aldrich has advertised such a chemically produced mixture in their catalogue. However, such a product did not offer interesting advantages.

One of the areas in which carbon black/polyaniline mixtures could offer an interesting technological advantage, could be the area of so the called 'supercapacitors', also

often called 'double layer' or 'redox capacitors'. This area provides the highest number of publications in which carbon and conductive polymers have been mixed or provided in form of a mixture.

For this purpose, principally two mixing procedures have been used:

- Simple mixing of powders of carbon black and polyaniline by one day of ball milling (US-Patent Application Publication No. 2002/0114128), or in other ways as disclosed in Journal of Power Sources 11, 2003, 185-190, and Journal of the Electrochemical Society, 148, 10, 2001, A1130-A1134, where a polyaniline or polythiophene derivative was mixed with carbon black powders. However, the specific means of mixing are not disclosed.
- Chemical or electrochemical polymerisation of different conductive polymers on the carbon black surface.

The latter method has been widely investigated in the patent and scientific literature. European Patent Application EP 1 329 918 reports about a negative electrode composite of carbon and polyaniline or polypyrrole in which the conductive polymer was electrochemically polymerised. The positive electrode was made from lead. The conductive polymer content was found to be optimal at 10-15% by weight. The capacitor made by using this material comprised an electrode combination including a positive non-polarisable and a negative polarizable electrode where the positive non-polarisable electrode was made from lead.

US Patent Application 2002/0089807 discloses an intrinsically conductive polymer directly polymerised on highly porous carbon black material by chemical or electrochemical means. The electrochemical polymerisation of polyaniline on a carbon aerogel generated from polyacrylonitrile is disclosed in Journal of Applied Electrochemistry 33, 465-473, 2003. That process is comparable to the electrochemical polymerisation of polyaniline on active porous carbon disclosed in Journal of Power Sources 117, 273-282, 2003 and lit. 10, Carbon 41, 2865-2871,

2003. In Conference Proceedings of ANTEC '98, Vol. 2, 1197ff, 1998, the authors report the polymerisation of polyethylenedioxythiophene on high surface area films of carbon on a platinum current collector support. In the same publication it was reported that polyaniline was cast from hexafluoro-isopropanol solutions on the same type of carbon films.

Electrochimica Acta Vol. 41, No. 1, 21-26, 1996 discloses various redox supercapacitors in symmetry or non-symmetric electrode configuration form in which various conductive polymers like polypyrrole or polythiophene derivatives have been used and polymerised on the substrate, in a comparable procedure as used in US-Patent 5,527,640, 1996, where polythiophene derivatives were polymerised on carbon substrates by electrochemical means.

As explained above, carbon black / conductive polymer mixtures have often been prepared for use in so-called "supercapacitors". Such capacitors are also often referred to as "double layer capacitors", "electrochemical" or "electrical double layer capacitors" or "redox capacitors" as well as sometimes "pseudo-capacity capacitors".

Conventionally, a double layer capacitor is an energy device in which two electrodes - at least one of which being obtained by coating a collector plate with a porous carbonaceous material having a high specific surface area between and above about 100 and 1000 m<sup>2</sup>/g to a collector plate - are disposed opposite to each other with a separator disposed therebetween. A voltage is impressed on the electrodes in the presence of an electrolyte solution, to generate an electrical double layer on at least one of the electrodes, and energy can be taken out therefrom. The structure of one kind of double layer capacitor using porous carbonaceous material as electrode material is, as disclosed in the specification and drawings of US Patent No. 5,150,283, classified into a type in which a pair of electrical double layer electrodes (each comprising a polarizable electrode joined to a collector plate) are wound and contained in a container, and a button type in which a pair of electrical double layer electrodes are laminated.

The wound type has a configuration in which a lead wire for giving off energy to the exterior is attached to a collector plate composed, for example, of an etched aluminum foil with a thickness of 20 to 50  $\mu\text{m}$ . The aluminum foil is coated with a paste composed of a powdered mixture prepared by admixing an active carbon powder with a desired binder and a desired conductive agent to form a conductive layer. A polarizable electrode composed of an active carbon layer consisting mainly of active carbon is formed on the conductive layer to obtain an electrical double layer electrode, and a pair of such electrical double layer electrodes is disposed opposite to each other with a separator disposed therebetween and is wound.

Alternatively, the electrical double layer electrodes are assembled by a method in which the polarizable electrode composed of the active carbon layers and the separator are sufficiently impregnated with an electrolytic solution containing an electrolyte dissolved therein under vacuum. The electrodes and the separator are inserted into a case made of aluminum or the like, and an opening portion of the aluminum case is sealed by use of a packing. Generally, this assembly is of a cylinder type.

On the other hand, the button type has a structure in which a polarizable electrode composed of an active carbon layer is formed on a disk-shaped sheet of a valve metal to obtain an electrical double layer electrode. A pair of such electrical double layer electrodes is disposed opposite to each other with an insulating separator therebetween, and this assembly is housed in a metallic container composed of two members. The two electrical double layer electrodes have their disk-shaped form sheets (or foils) of valve metal joined respectively to the inner sides of a bottom portion and a top cover portion of the metallic container, the bottom portion and the top cover portion are joined to each other while being hermetically sealed with an insulating ring packing at a circumferential edge portion thereof, and the inside of the container is filled with a nonaqueous electrolytic solution which is supplied sufficiently to the electrical double layer electrodes and the separator. As the

nonaqueous electrolytic solution, for example, a solution prepared by adding tetraethylammonium tetrafluoroborate to propylene carbonate is utilized.

There are several other configurations of double layer capacitors or redox capacitors (often called "supercapacitors") in use which will not be described here in detail.

There has been proposed a high-capacity electrochemical capacitor in which electrochemically active inorganic substances or organic (intrinsically) conductive polymers are used as electrode materials in combination with or in place of the above-mentioned porous carbonaceous material and in which electric power storage based on the formation of electrical double layers is utilized, in the same manner as in the ordinary electrical double layer capacitor using the above-mentioned porous carbonaceous material, and, simultaneously, electric power storage based on an oxidation-reduction potential attendant on oxidation-reduction reactions at both electrodes is utilized, to thereby achieve a high capacity.

For example, an electrochemical capacitor using ruthenium oxide as an electrode active substance (Physics Letters, 26A, p.209 (1968)) is evaluated to be the highest in performance of the electrochemical capacitors using an inorganic oxide known at present, and has been confirmed to have an energy density of 8.3 Wh/kg and an output density of 30 kW/kg.

In recent years, capacitors using other inorganic oxide mixture systems as electrode material have been investigated (J. Power Sources, vol. 29, p.355 (1990)).

Also, electrochemical capacitors using as electrode material (intrinsically) conductive polymers utilizing the oxidation-reduction characteristic of a  $\pi$ -conjugated system organic substance, other than the above-mentioned inorganic metal oxides, have been actively studied in recent years.

The electrochemical capacitors using the conductive polymers have been investigated in many research institutes, and, as to the characteristics, it has been reported that, for example, the use of polypyrrole as electrode material gives a capacity of 86 C/g and an energy density of 11 Wh/kg, the use of a mixture of polypyrrole and polythiophene gives a capacity of 120 C/g and an energy density of 27 Wh/kg, and the use of poly-3-(4-fluorophenyl)thiophene gives a capacity of 52 C/g and an energy density of 39 Wh/kg (J. Power Sources, vol. 47, p.89 (1994)).

Other than the above, in the type of utilizing intercalation for a high-capacity capacitor, generally, a layer structure substance ( $\text{TiS}_2$ ,  $\text{MoS}_2$ ,  $\text{CoO}_2$ ,  $\text{V}_6\text{O}_{13}$ ) is used as the electrode material. In this case, the device is in many cases assembled with asymmetric electrodes (Japanese Translations of PCT for Patent Nos. 2002-525864, 2002-542582).

Furthermore, composite materials prepared from the above-mentioned various electrode materials have been evaluated in some cases, to achieve much higher capacities, as compared to the conventional capacitors using only the porous carbonaceous material as an electrode material.

#### [Problems to be Solved by the Invention]

In spite of intensive research, no mixtures of carbonaceous materials and (intrinsically) conductive polymers are in industrial use. The reasons for this can mainly be found in a poor homogeneity of the mixture on the nanoscopic scale and a poor reproducibility of the properties due to the manufacturing processes (as described above) used.

On the other hand, the above-mentioned conventional electrical double layer capacitors using a porous carbonaceous material for polarizable electrodes are limited in capacity, though they are good in quick charge-discharge characteristics. For example, as described in Japanese Patent Laid-open No. 2003-338437, a capacitor

using active carbon (surface area: 650 m<sup>2</sup>/g) obtained by activation of a porous carbonaceous material has an electrostatic capacity of 18.3 F/g, which is lower than those of electrochemical capacitors.

As for the relationship between the electrostatic capacity and the specific surface area of the porous carbonaceous material, an electrostatic capacity of about 22 F/cc is obtained when the specific surface area is in the range of 1500 to 2600 m<sup>2</sup>/g, but, when the specific surface area is more than the range, the capacity is not increased any longer but tends to be decreased ("Electrical Double Layer Capacitors and Electric Power Storage Systems", The Nikkan Kogyo Shimbun, Ltd., p.9).

Other than the above, further examples of an electrode material using a porous carbonaceous material are disclosed in Japanese Patent Laid-open Nos. 2003-217982, 2003-81624, 2002-373835, and the like, but each of the examples gives a lower capacity, as compared to those of electrochemical capacitors, which matters particularly in use for automobiles and the like.

Besides, when using only the porous carbonaceous material for electrodes, it is necessary to add acetylene black or the like to the binder for the purpose of enhancing conductivity, but, even with such an addition, the interface resistance is very high, which constitutes a serious problem.

On the other hand, there are electrochemical capacitors using a conductive metal oxide as described above. Although such electrochemical capacitors have very high capacities as compared to the cases of using the above-mentioned porous carbonaceous material, the electrochemical capacitors are disadvantageous on the basis of production cost, since the metal of the metal oxide generally belongs to noble metals. For example, pseudo-capacity capacitors using a thin film of ruthenium oxide or indium oxide as electrode material have been reported to have an electrostatic capacity of 160 F/cc in an aqueous system, in the 5th International Seminar on Double Layer Capacitors and Similar Energy Storage Device held in Florida, USA, in 1995.



Besides, the electrochemical capacitors using a conductive polymer material as electrode material generally show a much higher capacity, as compared with those of capacitors using the above-mentioned porous carbonaceous material, but their characteristics may vary depending on the method of forming the electrodes. For example, where the conductive polymer is deposited on collector plates by an electrolytic polymerization method as described in Japanese Patent Laid-open Nos. Hei 6-104141 and Hei 6-104142, the capacitor has a capacity of 3.7 F and an internal resistance of 13.9  $\Omega$ , and the formation of electrodes by the electrolytic polymerization method generally involves a serious problem as to productivity thereof.

Therefore, many proposals have been made to use combinations of carbonaceous materials (mainly carbon black) and intrinsically conductive polymers. As described above, only two methods have been used to prepare such mixtures - a dry blending method of powders (by ball milling and the like) and the polymerisation of the conductive polymer in presence of the carbon black.

The method of synthesizing a conductive polymer by chemical oxidation polymerization and preparing a composite material from the conductive polymer and other conductive inorganic material is an apparently relatively simple technology for forming electrodes, and many examples of this technology have been made publicly available in recent years. However, its reproducibility is poor and therefore, no such mixtures are in practical use. Also, the problems to be overcome when trying to scale up and develop a reproducible process are huge.

For example, Japanese Patent Laid-open No. 2002-265598 discloses the use of a composite material of an organic conductive oligomer and an inorganic material as electrode material. In this case, however, the organic conductive oligomer cannot be expected to have a high conductivity, so that the inorganic material is required to have a very high conductivity in order to lower the internal resistance.

Besides, as a similar case of using a conductive polymer material, a method of polymerizing the conductive polymer in a porous carbonaceous material (Japanese Patent Laid-open No. 2001-210557) cannot produce a conductive polymer with a high conductivity but leads to an increase in the internal resistance of the capacitor.

As has been described above, there have hitherto been known a large number of examples of a capacitor using an electrochemically active conductive polymer or inorganic oxide. However, the use of the inorganic material leads to a problem as to production cost, whereas the use of the conductive polymer involves difficulties in control of conductivity, control of particle size and reproducibility, resulting in that the capacitors cannot show sufficient performance.

It was therefore the object of this invention to create a carbon material / conductive polymer mixture which can be manufactured in a reproducible manner and shows superior performance values especially for the manufacture of supercapacitors.

[Detailed description of the invention]

Most surprisingly, a mixture of (intrinsically) conductive polymer(s) with carbon materials like carbon black, graphite, carbon nanotubes or fullerenes, can reproducibly be manufactured, if the conductive polymer(s) is (are) provided in colloidal form and the colloidal form of the conductive polymer is mixed with the carbon material.

Thus, in a first aspect, the present invention relates to composition capable of forming a coating and comprising a mixture of a conductive polymer in colloidal form and carbon.

In a second aspect, the present invention relates to a method for the manufacture of a composition according to the first aspect, said method comprising dispersing the conductive polymer and carbon, and optionally additives in a liquid dispersion medium and optionally drying the liquid dispersion after application on a substrate.

In a third aspect, the present invention relates to a composite material comprising the composition according to the first aspect or the composition obtained by the method of the second aspect in the form of a coating on a substrate.

In a fourth aspect, the present invention relates to an electrical or electronic article comprising the composition according to the first aspect or the composite material according to the third aspect.

Preferred embodiments of the present invention are disclosed in the dependent claims.

Preferably, the carbon materials of the present invention are selected from the group consisting of graphite, carbon black, nanotubes and fullerenes. The carbon materials of the present invention are also referred to as 'carbon-based materials' since they do not necessarily consist exclusively of the chemical element carbon but may also comprise other elements such as hydrogen, oxygen, nitrogen and sulphur. For example, carbon black may contain 0.3-1.3 % H, 0.1-0.7 % N, and 0-0.7 % S. Depending on the mode of manufacture, the oxygen content of carbon black may amount to 0-1.5 % for furnace and thermal type carbon blacks, up to 5 % for gas and channel type carbon blacks, and up to 15 % for carbon black post-treated with oxygen. Further, condensed aromatics may be present at the surfaces of the material.

The term "(intrinsically) conductive polymers" (ICP) refers to organic polymers which have (poly)-conjugated  $\pi$ -electron systems (e.g. double bonds, aromatic or heteroaromatic rings or triple bonds). They can exist in various states, each described by different empirical formulae, which can generally be converted essentially reversibly into one another by (electro)chemical reactions such as oxidation, reduction, acid/alkali reaction or complexing. These reactions are also occasionally known as "doping" or "compensation" in the literature, or can be regarded as "charging" and "discharging" in analogy with the electrochemical processes in batteries. At least one of the possible states is a very good conductor of electricity, e.g. has a conductivity of more than 1 S/cm (in pure form), so one can speak of

intrinsically conductive polymers. These forms of the ICP are generally regarded, as poly-radical cationic or anionic salts. A good overview of the (intrinsically) conductive polymers synthesised to date with a chemical structure suitable for the present objective, can be found in Synthetic Metals, Issues 17, 18 and 19 (1987), and in Synthetic Metals (in press), Proceedings of the ICSM '88 (Santa Fé).

The chemical nature of the conductive polymers of the invention is not particularly limited, and examples thereof include polyaniline, polyaniline derivatives, polythiophene, polythiophene derivatives, polypyrrole, polypyrrole derivatives, polythianaphthene, polythianaphthene derivatives, polyparaphenylene, polyparaphenylene derivatives, polyacetylene, polyacetylene derivatives, polydiacetylene, polydiacetylene derivatives, polyparaphenylenevinylene, polyparaphenylenevinylene derivatives, polynaphthalene, and polynaphthalene derivatives, polyisothianaphthene (PITN), polyheteroarylenevinylene (ParV), in which the heteroarylene group can be e.g. thiophene, furan or pyrrol, polyphenylene-sulphide (PPS), polyperinaphthalene (PPN), polyphthalocyanine (PPhc) etc., and their derivatives (formed for example from monomers substituted with side chains or groups), their copolymers and their physical compounds. The method for polymerizing the conductive polymers is not particularly limited, and the usable methods include electrolytic oxidation polymerization, chemical oxidation polymerization, and catalytic polymerization. The polymer obtained by the polymerizing method as mentioned is neutral and is not conductive per se. Therefore, the polymer is subjected to p-doping or n-doping to be transformed into a conductive polymer. The substance used for the doping is not particularly limited; generally, a substance capable of accepting an electron pair, such as a Lewis acid, is used. Examples include hydrochloric acid, sulfuric acid, organic sulfonic acid derivatives such as parasulfonic acid, polystyrenesulfonic acid, alkylbenzenesulfonic acid, camphorsulfonic acid, alkylsulfonic acid, sulfosalicylic acid, etc., ferric chloride, copper chloride, and iron sulfate.

It is not critical in the practice of the present invention in which form the carbon based material is provided. The colloidal form of the conductive polymer can be mixed with the carbon based material provided in (dry or wet) powder form, in form of a porous film or composite or also in form of a colloidal formulation. The carbon can be pretreated (e.g. by ball milling), predispersed in water, organic solvent or another medium or shaped in form of fibres, nanofibres, films or porous shapes or membranes. The colloidal form of the conductive polymer may be added to the carbon based material or vice versa.

For instance, the carbon powder can be added to a colloidal dispersion of the conductive polymer. Alternatively, the conductive polymer dispersion can be added to the carbon by coating a carbon substrate with the colloidal dispersion of the conductive polymer.

The colloidal form of the conductive polymer can be manufactured in various different ways which are basically known and state of the art. See for example EP-A-0 329 768 and German patent application No. 10 2004 003 784. According to the present invention, polymer particles having a size of smaller than 500 nm are considered to be colloidal. The size of the particles can be determined by scanning or transmission electron microscopy, or by using laser Doppler technology. Another method of manufacture of the polymer is its polymerisation in the presence of surface active agents so that the resulting polymer remains in colloidal form in the reaction medium. For the subsequent use as colloidal dispersion, one would purify the dispersion by ion exchange and / or membrane filtration.

Alternative methods are procedures by which the conductive polymer is polymerised, then neutralised (compensated) to the non-conductive form and mixed with solvents resulting in a fine colloidal dispersion (often considered to be a "solution"). This colloidal dispersion of the non-conductive form of the conductive polymer can be mixed with the carbon based material and the resulting mixture can be rendered

conductive by adding appropriate materials ("dopants") like Broenstedt acids, Lewis acids or oxidants (like iodine,  $\text{FeCl}_3$  or else).

Another preferred way is to polymerise the conductive polymer so that it precipitates from the reaction medium, can be filtered, washed and afterwards dispersed. This manner of polymerisation is described in EP-B-0 329 768. Polymers obtained by that process are commercially available from Ormecon GmbH, Germany. See in particular Example 1 and the description at page 8, lines 7-22 of EP-B-0 329 768.

The most preferred way is to use conductive polymer dispersions manufactured as described in German patent application No. 10 2004 003 784, filed January 23, 2004. See in particular claim 6 and Examples 1 to 8 therein. The process of German patent application No. 10 2004 003 784 allows one to prepare colloidal conductive polymers having a conductivity of above 100 S/cm which is advantageous for many applications. In the process, the conductive polymer is in general first polymerised according to the polymerization method of EP-B-0 329 768 so that the primary particles have a diameter of less than 500 nm, and the resulting powder is dispersible. According to one embodiment, this powder is then dispersed in a first dispersion step, followed by a conditioning step and a second dispersion step. Finally, the dispersion is formulated so that the colloidal dispersion meets the various following processing and applications requirements (viscosity, stability, drying time and temperature, and others). Suitable dispersion techniques are described in German patent application No. 10 2004 003 784 and include, for example, the use of an ordinary ball mill, a planetary ball mill, a homogenizer, or an ultrasonic dispersing apparatus.

The conductive polymer can be mixed with the carbon based material after the second dispersion step (which usually results in a highly viscous paste form) or after the final formulation step (which usually results in a less viscous liquid).

The carbon based materials preferably used for this invention are carbon nanotubes, nanofibres, fullerenes, carbon black and the like. These materials tend to agglomerate and are usually not easily impregnated with liquids or colloidal dispersions.

Because it is advantageous for the later use of the carbon / conductive polymer mixtures, the surface of the carbon based materials should be accessible for the colloidal conductive polymer. Therefore it is preferred to provide the carbon based material in form of highly porous structures, fine powders or colloidal dispersions. If dry powders are used, it is preferred to pretreat these powders, e.g. by ball milling. Ball milling can be carried out in planetary mills or satellite ball mills.

The carbon based materials can also be dispersed in presence of the colloidal dispersion of the conductive polymer, e.g. in ball mills.

It is also preferred to manufacture a dispersion of the carbon based material in water or organic solvents (with or without using surface active agents, tensides and other materials). Such a dispersion can be made in ball mills, pearl mills, ultrasonic dispersion apparatuses, 2- or 3-roll mills or other state-of-the-art dispersion machines.

In order to attain a highly dispersed state, the dispersing process can be carried out by use of an ordinary ball mill, a planetary ball mill, a homogenizer, or an ultrasonic dispersing apparatus. Where the planetary ball mill is used, stirring is preferably carried out for at least 30 min. Stirring for a long time may be associated with processing problems such as a rise in the temperature of the solvent. Therefore, cooling is needed in the case of stirring for more than 1 hour.

It is preferred to produce a paste of the carbon based material which can be mixed with the conductive polymer colloid without, before, during or after further dilution and formulation.

The relation between carbon and conductive polymer may widely vary. For example, the weight ratio of the conductive polymer to carbon may range from 1 : 50 to 50 : 1 or 1 : 50 to 50 : 2.

The final mixture of carbon and conductive polymer may contain additives like surface active agents or stabilisers, and other materials.

The selection of the carbon based material to be combined with the colloidal conductive polymer is only limited by availability, price and suitability for the final application. Mixtures of different carbon based materials, like active carbon black and conductive carbon black may be advantageous in certain applications.

The same applies to the conductive polymer. Although it is not critical to the process or application requirements of this invention, availability and price may limit the selection of conductive polymers. Generally, all of the above described conductive polymers are useful in the practice of the present invention. Preferred are polyanilines and polythiophenes and their derivatives, in particular polyaniline and its copolymers and derivatives.

Carbon / conductive polymers according to this invention can be used in conductors, energy stores, sensors, switches, condensers, capacitors and supercapacitors, double layer capacitors and redox capacitors

For the application of carbon / conductive polymers according to this invention for the storage of (electrical) energy, e.g. in capacitors, active carbon black and polyanilines or polythiophenes and their derivatives are preferred. Conductive carbon black may be used as an additional component as well.

A preferred application of the compositions according to the invention is their use as electrode material in high-capacity capacitors. Such capacitors are often called "super-capacitors" or "double layer capacitor" ("DLC"), also sometimes (referring to some



specific mechanism in certain forms of such capacitors) "redox capacitors". In the following, this application will be referred to as "DLC".

One preferred embodiment of the invention is the manufacturing of materials for electrodes for a high-capacity capacitor comprising a porous carbonaceous material or a valve metal, at least one of the electrodes being formed from a composition containing an electrochemically active substance. Surprisingly, by combining the carbonaceous material with the conductive polymer in colloidal form, it is possible to strongly bind the porous carbonaceous material and to form a stable electrode or electrodes. While in the prior art a non-conductive organic binder has been used as a binder for binding the porous carbonaceous material, the conductive material according to the invention makes it possible to form the electrode without using a binder. In addition, the adhesive force for adhesion to a collector plate can be sufficiently secured by controlling the particle diameter. Moreover, the performance data indicate the superior formation of an electrical double layer, and the material according to the invention can easily impregnated with the electrolyte solution.

Although it is not critical for the scope of this invention, usually, for the use as electrode material, the mixture contains a higher amount of (active) carbon than conductive polymer.

The conductivity of the conductive polymer contributes particularly to a reduction in the equivalent series resistance of the capacitors. While it is a general practice to add acetylene black or the like as a conductivity-imparting material in the case of an electrode for a capacitor which uses a porous carbonaceous material, the material according to the invention is characterized in that, by providing a conductive polymer with conductivity of 100 S/cm or more, the conductive polymer by itself has the function as a conductivity-imparting material, making it unnecessary to especially add a conductivity-imparting material. Since it is unnecessary to add a conductivity-imparting material, it is possible to enlarge the capacity of the actual product. In this case, the conductivity of the conductive polymer is at least 100 S/cm, preferably 200

S/cm or more, and more particularly 500 S/cm or more. The conductivity is preferably as high as possible, since it has a great influence on the internal resistance of the capacitor to be made.

In a further embodiment of the invention, the conductive polymer is dispersed in water or an organic solvent. Particularly, the composition specified conditionally can be used separately for the case where the electrolyte in the capacitor is water and for the case where the electrolyte is an organic solvent. Especially where the electrolytic liquid is water, the use of the conductive polymer dispersed in an organic solvent is advantageous to the stability of the electrode.

In another embodiment of the invention, an electrode composition for a capacitor is provided, wherein the conductive polymer dispersion element contains solid components in a concentration of not more than 20 wt%. Particularly at the time of forming the electrodes, the concentration of the solid components greatly influences the productivity and cost. However, the solid components in the conductive polymer dispersion element influences the stability of the dispersion, too. Therefore, the concentration of the solid components is not more than 20 wt% where production cost is taken into account, and the concentration is not more than 10 wt%, preferably not more than 5 wt%, where the stability of the dispersion is also taken into account.

In still another embodiment of the invention, it is preferred to focus on the specific surface area of the porous carbonaceous material. A sufficient capacity of the DLC can be obtained by use of a carbonaceous material with a specific surface area of not less than 100 m<sup>2</sup>/g, preferably 500 m<sup>2</sup>/g, more preferably 1000 m<sup>2</sup>/g. Such a porous carbonaceous material is dispersed in the conductive polymer dispersion element.

The amount of the porous carbonaceous material used can widely vary. It is necessary to add the porous carbonaceous material in an amount of at least 5 wt%, in relation to the weight of solid polyaniline.

The addition amount is closely related to the charge-discharge rate of the capacitor. It is desirable to add the porous carbonaceous material in a larger amount in the case of devices for quick charge-discharge, and to add the conductive polymer in a larger amount in the case of devices for slow charge-discharge. Thus, according to the present invention, it suffices to control the addition amount according to the device requirements. In order to obtain a capacitor with a capacity of at least greater than that of a capacitor using solely the porous carbonaceous material, it is necessary to add the conductive polymer in an amount in terms of solid components of not less than 5 wt%. In order to obtain a sufficient oxidation-reduction performance, it is necessary to add the conductive polymer in an amount in terms of solid components of preferably not less than 10 wt%.

In still another embodiment of the invention, the materials are used for electrodes for a DLC, said electrodes being disposed opposite to each other with the separator disposed therebetween, wherein both of the two electrodes are composed of the same kind of conductive polymer. This configuration has the merit of providing a very inexpensive device, in consideration of the productivity thereof.

In still another embodiment of the invention, the materials are used for electrodes for a DLC, wherein the compositions used to form the two electrodes disposed opposite to each other with the separator disposed therebetween comprise different kinds of conductive polymers. This is based on the consideration that many of organic conductive polymers show different activities under a positive voltage and under a negative voltage. In other words, a method of obtaining a high-capacity device resides in using a conductive polymer material which shows the highest electrical activity when the voltage is swept.

In still another embodiment of the invention, the materials are used for electrodes for a DLC, wherein one of the two electrodes comprises a porous carbonaceous material, and the other comprises a porous carbonaceous material/conductive polymer

composite according to the invention. This configuration shows a surprisingly high capacity.

In still another embodiment of the invention, the materials are used for electrodes for a DLC, wherein one of the two electrodes comprises a metal oxide, and the other comprises a porous carbonaceous material/conductive polymer mixture according to the invention. This configuration is effective particularly in the case of obtaining a high capacity by utilizing the oxidation-reduction characteristics of the metal oxide electrode. Particularly desirable examples of the material of the metal oxide electrode are ruthenium oxide and indium oxide.

Moreover, where the inorganic material/conductive polymer composite body in the present invention is formed into a film or other molded product, it is possible to add a stabilizer, a light stabilizer, a filler, a binder, a conductivity-imparting agent and the like, as required.

The effects of using carbon / colloidal conductive polymer mixtures as electrode materials in DLC are quite surprising. Compared with such types of capacitors manufactured by polymerising a conductive polymer in presence or on the surface of (active) carbon black used in DLC, the capacitors manufactured using the mixtures according to this invention exhibit significantly better properties. However, it is noted that the scope of this invention is not limited by the specific properties of the DLC manufactured according to the invention. In other words, DLCs which may show a lower performance are also considered to be within the scope of this invention.

The capacity may range from, for example, over 50 to well over 200 F/g, the power density may range, for example, from about 1,000 to several 1,000 W/kg, and the energy density ranges, for example, from about 10 and well over 100 Wh/kg.

As can be seen in Figure 6 and Figure 11 (Ragon plots), this set of characteristics positions the DLC according to the invention outside of the presently accessible range

of power and energy density, significantly higher than DLCs (supercapacitors) and even Lithium ion batteries of the present state of the art.

Some of the reasons for this surprising effect may be – without intending to be limited by this explanation – (a) that there is no need any more for adhesive or film forming additives (in order to form an adhering electrode material layer on the surface of the metallic collector) (b) a lower internal resistance and (c) the additional possibility of redox reactions, all together increasing the accessible surface (accessible for charge carriers and their exchange), the wetting of the electrolyte on the electrode material and the energy density (charge per weight).

[Mode for Carrying Out the Invention]

## **Examples**

### **Example 1: Production of Porous Carbonaceous Material/Conductive Polymer Composite Electrodes**

Pulverized coconut shell active carbon (not yet activated) in an amount of 18 g was dispersed in 90.9 g of a conductive polymer dispersion (ORMECON 7301-026-002, a dispersion of conductive polyaniline in xylene; solid components: 2.2%), and the mixture was stirred by a planetary ball mill for 60 min. To the porous carbonaceous material/conductive polymer slurry, 57.3 g of xylene was further added, and the mixture was stirred by a stirring motor for 30 min, to prepare a porous carbonaceous material/conductive polymer dispersion.

Electrodes for a capacitor need collector plates adapted to the electrolytic solution used in the capacitor. In this example, platinum plates were used for electrodes when using aqueous sulfuric acid solution (1 mol/l) as the electrolytic solution, and aluminum was used for the electrodes when using propylene carbonate as the electrolytic solution. Where platinum is used for the collector plates in the actual

electrodes for the capacitor, the surfaces of platinum plates are first scratched with a file, and are then coated with a fixed amount of the porous carbonaceous material/conductive polymer dispersion prepared as described above. Then, the coated platinum plates are placed in a high-temperature tank at 100°C for 1 hour to sufficiently remove the organic solvent. The electrodes thus formed were subjected to weighing of the electrode active substance, and were used directly as the electrodes for the capacitor.

When using a 1 mol/l solution of tetraethylammonium tetrafluoroborate in propylene carbonate as an organic solvent based electrolytic solution, aluminum plates were used as collector plates. The aluminum collector plates were scratched by a file and coated with a fixed amount of the porous carbonaceous material/conductive polymer dispersion, in the same manner as in the case of using the platinum plates as the collector plates. Thereafter, drying was conducted at 100°C for 24 hours, to sufficiently remove moisture.

The components of the dispersion used for forming the porous carbonaceous material/conductive polymer composite electrodes are shown in the following Table 1.

**Table 1**

	Conductive polymer	Dispersion medium	Specific surface area of active carbon (m <sup>2</sup> /g)	Solid component ratio C:ICP
Experiment 1	Polyaniline	Xylene	1100	9:1
Experiment 2	Polyaniline	Xylene	1100	8:2
Experiment 3	Polyaniline	Xylene	1100	1:9
Experiment 4	Polyaniline	Xylene	1600	9:1
Experiment 5	Polyaniline	Xylene	1600	8:2
Experiment 6	Polyaniline	Xylene	1600	1:9
Experiment 7	Polyaniline	Water	1600	8:2
Experiment 8	PEDOT	Water	1600	8:2

**Example 2: Production of Capacitor Cell**

Each electrode plate with the electrode active substance in close contact therewith, as prepared in Example, 1 is blanked into circular disks with a diameter of 1 cm, to prepare two electrodes. A glass fiber filter is blanked into a circular shape with a diameter of 1.5 cm and used as a separator. Further, an aqueous 1 M solution of sulfuric acid is used as an electrolytic solution in the case of an aqueous system. A 1 M solution of tetraethylammonium tetrafluoroborate in propylene carbonate is used as an electrolytic solution in the case of an organic solvent system.

**Evaluation of Capacitor Characteristics**

Measuring Instruments: The internal resistance of each capacitor cell was measured by use of an impedance analyzer YHP 4192A, available, from Hewlett-Packard Development Company, L.P. Measurements in a charge-discharge test were carried out by use of a TOYO System TOSCAT-3100U, available from TOYO SYSTEM, Co., LTD.

The results of the charge-discharge test of capacitors using the electrode active substances shown in Experimental Example 1 are shown as Examples in the following Table 2.

**Table 2**

Test No.	Electrode active substance	Dispersion	Charging rate (mA/cm <sup>2</sup> )	Charging capacity (F/g)	Discharging rate (mA/cm <sup>2</sup> )	Discharging capacity (F/g)
1	Experiment 1	Sulfuric acid solution	10	250	10	250
2	Experiment 1	Sulfuric acid solution	5	240	5	240
3	Experiment 1	Sulfuric acid solution	1	220	1	220

4	Experiment 2	Sulfuric acid solution	10	250	10	250
5	Experiment 2	Sulfuric acid solution	5	220	5	220
6	Experiment 2	Sulfuric acid solution	1	170	1	170
7	Experiment 2	TEATFB	10	90	10	90
8	Experiment 2	TEATFB	5	100	5	100
9	Experiment 2	TEATFB	1	150	1	150
10	Experiment 4	TEATFB	10	40	10	40
11	Experiment 4	TEATFB	5	170	5	170
12	Experiment 4	TEATFB	1	170	1	170
13	Experiment 5	TEATFB	10	140	10	140
14	Experiment 5	TEATFB	1	200	1	200

The results of evaluation of internal resistance are shown in Figure 1 and Figure 2. Measurement was conducted by use of a capacitor cell formed by using the electrodes using the dispersion elements of Experiment 2 and Experiment 5, and using a 1 M sulfuric acid solution as the electrolytic solution.

Fig. 1 shows the internal resistance of the capacitor (impedance characteristics) using the porous carbonaceous material with a specific surface area of 1600 m<sup>2</sup>/g.

Fig. 2 shows the internal resistance of the capacitor (impedance characteristics) using the porous carbonaceous material with a specific surface area of 1100 m<sup>2</sup>/g.

### Comparative Example

A mixture of an electrode material including 60 wt% activated carbon (specific surface area of about 1600cm<sup>2</sup>/g), 30 wt% carbon black, and 10 wt% tetrafluoroethylene as binder was prepared. For film formation methanol was added to this mixture. An electrode sheet (10mm width, 10mm length, 3 mm thickness) was



prepared by using a rolling method. This sheet was dried at 250°C at otherwise ambient conditions.

Charge-discharge test were then carried out using the same method as described above in Example 2. In this case, the charge capacity was about 4.7 F/g only.

### **Example 3**

#### **1) Agent and apparatus:**

BELLFINE AP, which is produced by KABEBO LTD, was used as active carbon material.

D1005W (dispersion of polyaniline in water) and 7201-026-001(dispersion of polyaniline in xylene), both available from Ormecon GmbH, Ammersbek, Germany were used as conductive polymers in colloidal form.

Charge-discharge test was carried out by the use of TOCAT-3100U battery testing unit (TOYO SYSTEM, CO., LTD.). Impedance property was measured using 4192A LF IMPEDANCE ANALYZER (HEWLETT PACKARD).

#### **2) Preparation of a mixture of conductive polymer and active carbon:**

The mixture was prepared by using the following procedure. 8.0g of the active carbon and 90.9g of ORMECON 7201-026-001(solid :2.20%) were mixed using a Satellite ball mill for 60min. This mixture was used as electrode material. Electrode films were made by applying a 200 µm coating of the material on an aluminum plate. The wet film was dried at 100 °C under ambient conditions.

**Results:**

1: Composition obtained from organic (xylene) polyaniline dispersion (7201-026-001)

Charge-discharge testing was carried out using the thin film having a mixture of 7201-026-001 and BELLFINE AP. Charge-discharge properties were determined at  $0.5\text{mA/cm}^2$ ,  $1\text{mA/cm}^2$ , and  $5\text{mA/cm}^2$ . Initial capacities under these conditions are listed in Table 3.

**Table 3:** Initial capacity of ICP based capacitor.

Current density (mA/cm <sup>2</sup> )	Capacity (F/g)	Power density (W/kg)	Energy Density (Wh/kg)
0.5	254.9	1090	141.6
1.0	237.4	1805	131.9
5.0	211.3	5076	117.4

Both power density and energy density were excellent. This capacity is bigger than that of a capacitor made from BELLFINE carbon only. For the 'BELLFINE only' capacitor, the capacity is about only 30 F/g. Results of cycle tests are indicated in Figure 3, Figure 4, and Figure 5.

Fig.3 shows the result of a cycle test of capacitor at a current density of 0.5mA and a voltage of 2.0V using a propylene carbonate electrolyte including 0.01M tetraethylammonium tetrafluoroborate.

Fig.4 shows the result of a cycle test of capacitor at a current density of 1.0mA and a voltage of 2.0V using a propylene carbonate electrolyte including 0.01M tetraethylammonium tetrafluoroborate.

Fig.5 shows the result of a cycle test of capacitor at a current density 5.0mA and a voltage of 2.0V using a propylene carbonate electrolyte including 0.01M tetraethylammonium tetrafluoroborate

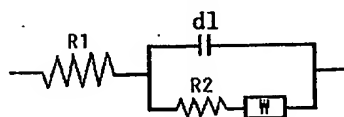
After the 4000cycle test, the capacitance decreased to about 1/3 of the initial value. Capacitance after 4000cycles is listed in Table 4.

**Table 4:** Capacity of ICP Based Capacitor. After 4000cycles

Current density (mA/cm <sup>2</sup> )	Capacity (F/g)	Power density (W/kg)	Energy density (Wh/kg)
0.5	74	1099	41.2
1.0	50	2051	27.9
5.0	73	5188	41.0

A Ragon-plot of this cycle test data is shown in Fig.6. It can be seen that the capacitance of the evaluated capacitor is higher than that of conventional ones. Specifically, energy density decreases with an increasing number of cycles. Clearly, however, power density hardly changes as compared to the initial value. In the case of 5mA/cm<sup>2</sup> condition, the characteristic got worse quickly after about 4500 cycles.

In order to understand the resistance components, impedance characteristics were measured. Measurement of impedance was carried out in the range from 5Hz to 13MHz. The interface action of ion species, however, cannot be seen in this frequency area. First, the resistance icomponent was checked by a Cole-Cole plot. Then change of each resistance was conformed by comparing with the Cole-Cole plot of initial resistance component with that of resistance component after 4000 cycles. In the case of this capacitor, equivalent circuit was depicted as follow;



In the case of a capacitor, R1 is a solution resistance which is in series with electrode interfacial impedance, and R2 is the Faradaic resistance of electrode process at reversible potential. This value is influenced by the resistance of conducting polymer. The Cole-Cole plot is shown in Fig.7.

R1 and R2 values are listed in Table 5.

**Table 5:** Resistance of capacitor using Conductive polymer and active carbon mixture as electrode.

Resistance	0.5mA/cm <sup>2</sup> R1( $\Omega$ )	0.5mA/cm <sup>2</sup> R2( $\Omega$ )	5.0mA/cm <sup>2</sup> R1( $\Omega$ )	5.0mA/cm <sup>2</sup> R2( $\Omega$ )
Initial	27	18	18	18
4000cycle	18	430	18	200

The above data shows that, although R1 did not change with both current density, R2 changed drastically with 4000 cycles. It is assumed that this change was dependent on the change of conductive polymer resistance such as elimination of dopant, oxidation of conductive polymer, decomposition of polymer and so on.

## 2. Water dispersed polyaniline (D1005W)

Charge-discharge test was carried out by the use of thin film on Al plate, which was made from mixture of D1005W and BELLFINE AP. Charge-discharge tests were performed at 0.5mA/cm<sup>2</sup>, 1mA/cm<sup>2</sup>, and 5mA/cm<sup>2</sup>. Initial capacities under these conditions are listed in Table 6.

**Table 6:** Initial capacity of ICP based capacitor.

Current density (mA/cm <sup>2</sup> )	Capacity (F/g)	Power density (W/kg)	Energy Density (Wh/kg)
0.5	185.1	1225	102.8
1.0	66.07	1549	36.7
5.0	106.7	12769	59.3

Energy density of capacitor using D1005W was less than that of capacitor using 7201-026-001. Results of Cycle test are shown in Fig.8 , Fig 9, and Fig 10.

Fig. 8 shows the result of a cycle test of capacitor at a current density of 0.5mA and a voltage of 2.0V and using a propylene carbonate electrolyte including 0.01M tetraethylammonium tetrafluoroborate

Fig 9 shows the result of a cycle test of capacitor at a current density of 1.0 mA and a voltage of 2.0V and using a propylene carbonate electrolyte including 0.01M tetraethylammonium tetrafluoroborate.

Fig. 10 shows the result of a cycle test of capacitor at a current density of 5.0 mA and a voltage of 2.0V and using a propylene carbonate electrolyte including 0.01M tetraethylammonium tetrafluoroborate.

In the case of 1.0mA/cm<sup>2</sup>, change of capacitance was not observed in the cycle test. In the case of 5.0mA/cm<sup>2</sup>, however, a rapid reduction of capacitance was observed at 1000 cycles. The capacity data after 4000 cycles are listed in Table 7.

Table 7: Capacity of ICP based capacitor of 4000cycles after .

Current density (mA/cm <sup>2</sup> )	Capacity (F/g)	Power density (W/kg)	Energy Density (Wh/kg)
0.5	55.1	1233	30.6
1.0	63.1	1311	35.1
5.0	19.4	10621	10.7

A Ragon-plot of the cycle test data are shown in Fig. 11.

From the data it is apparent that energy density decreases with an increasing number of cycles. However, power density hardly changes as compared to the initial value. At a current density of 1.0mA/cm<sup>2</sup>, power density and energy density were very stable for about 4000 cycles. At 5.0mA/cm<sup>2</sup>, power density got worse quickly after about 1000 cycles.

In order to analyse the resistance component, impedance measurements was carried out using impedance analyzer. Results of the impedance measurement are shown in Fig. 12, which is a Cole-Cole Plot of capacitor using mixture of conductive polymer and active carbon as electrode.

R1 and R2 values from the Cole-Cole plot are listed in Table 8.

**Table 8:** Resistance of capacitor using Conductive polymer and active carbon mixture as electrode.

Resistance	0.5mA/cm <sup>2</sup> R1( $\Omega$ )	0.5mA/cm <sup>2</sup> R2( $\Omega$ )	1.0mA/cm <sup>2</sup> R1( $\Omega$ )	1.0mA/cm <sup>2</sup> R2( $\Omega$ )	5.0mA/cm <sup>2</sup> R1( $\Omega$ )	5.0mA/cm <sup>2</sup> R2( $\Omega$ )
Initial	27	150	7.4	7.8	27	37
4000cycle	130	700	65	150	130	520

This R2 resistance component is bigger than that of 7201-026-001 based capacitor's R2. Without being bound by theory, it is assumed that the difference of R2 resistance between D1005W based capacitor and 7201-026-001 based capacitor can be attribute to difference of conductivity between D1005W and 7201-026-001. Both of R1 and R2 resistance increased drastically after 4000 cycles.

#### [Effects of the Invention]

As has been described above, the present invention provides an electrical double layer capacitor in which a pair of polarizable electrodes each comprising a solid electrolyte of a conductive polymer contained in an active carbon layer are opposed to each other with a separator disposed therebetween. Particularly, by enhancing the particle diameter and conductivity of the conductive polymer, it is possible to obtain an electrical double layer capacitor having a low internal resistance and a high capacity.